

Supplementary material for:

Evolution of Non-Metallic Inclusions Through Processing in Ti-V Microalloyed 316L and Al-V Microalloyed 17-4PH Stainless Steels for Hipping Applications

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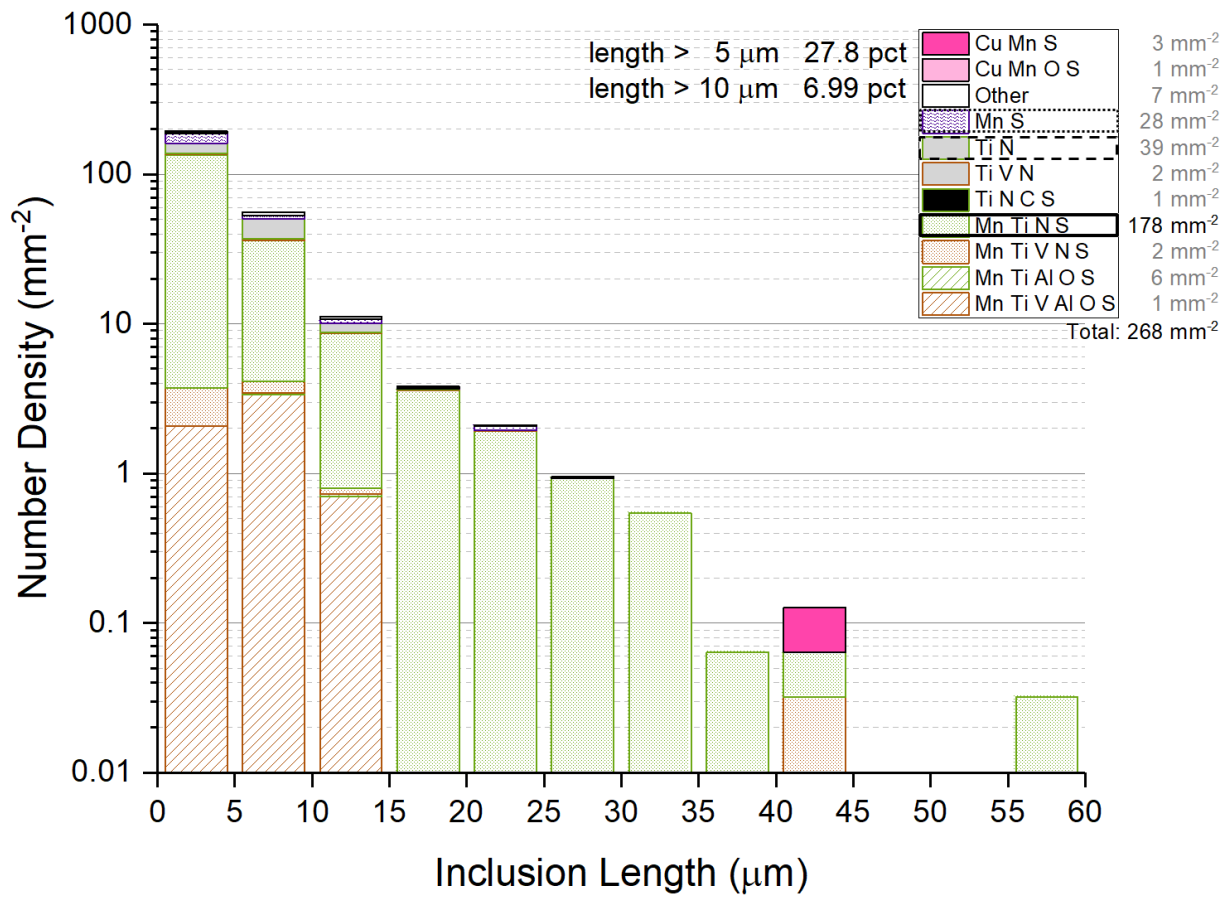
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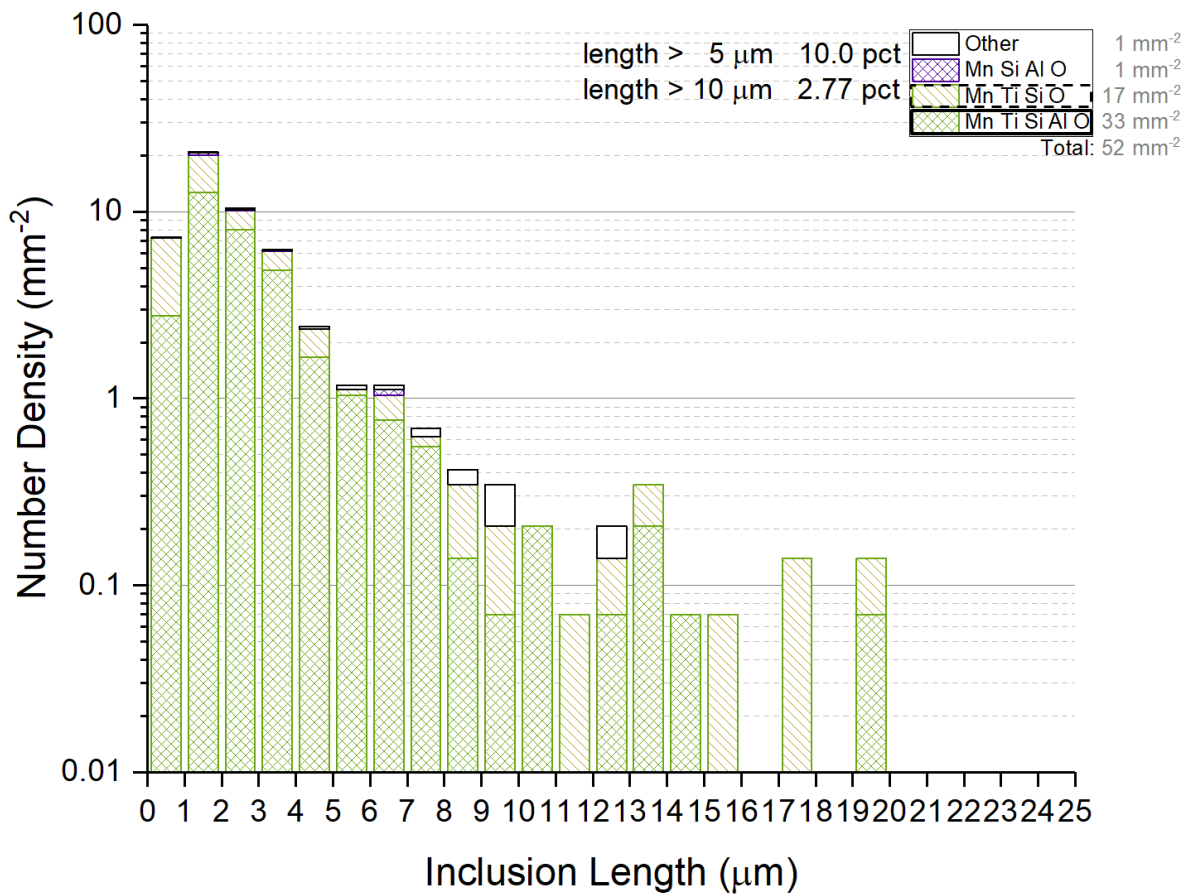
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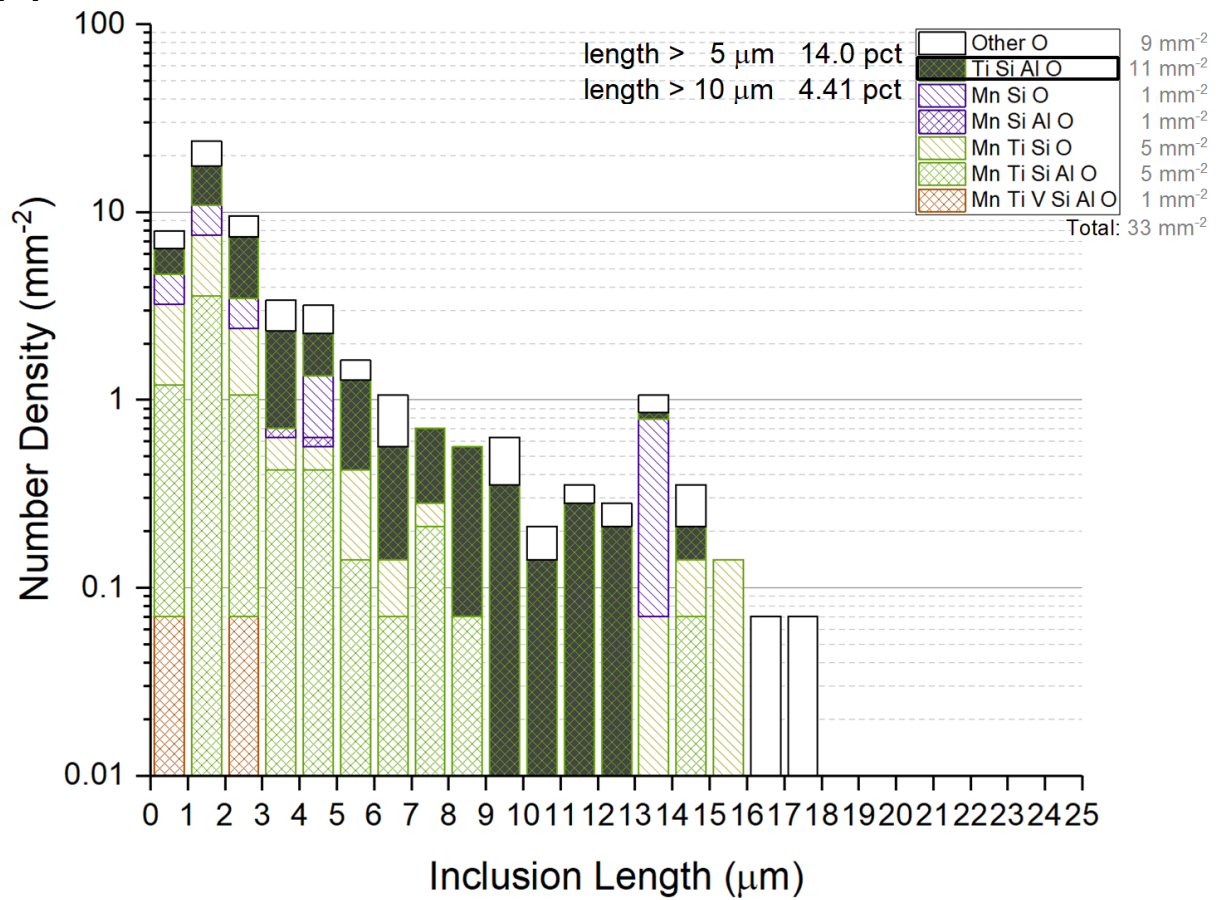
(a)



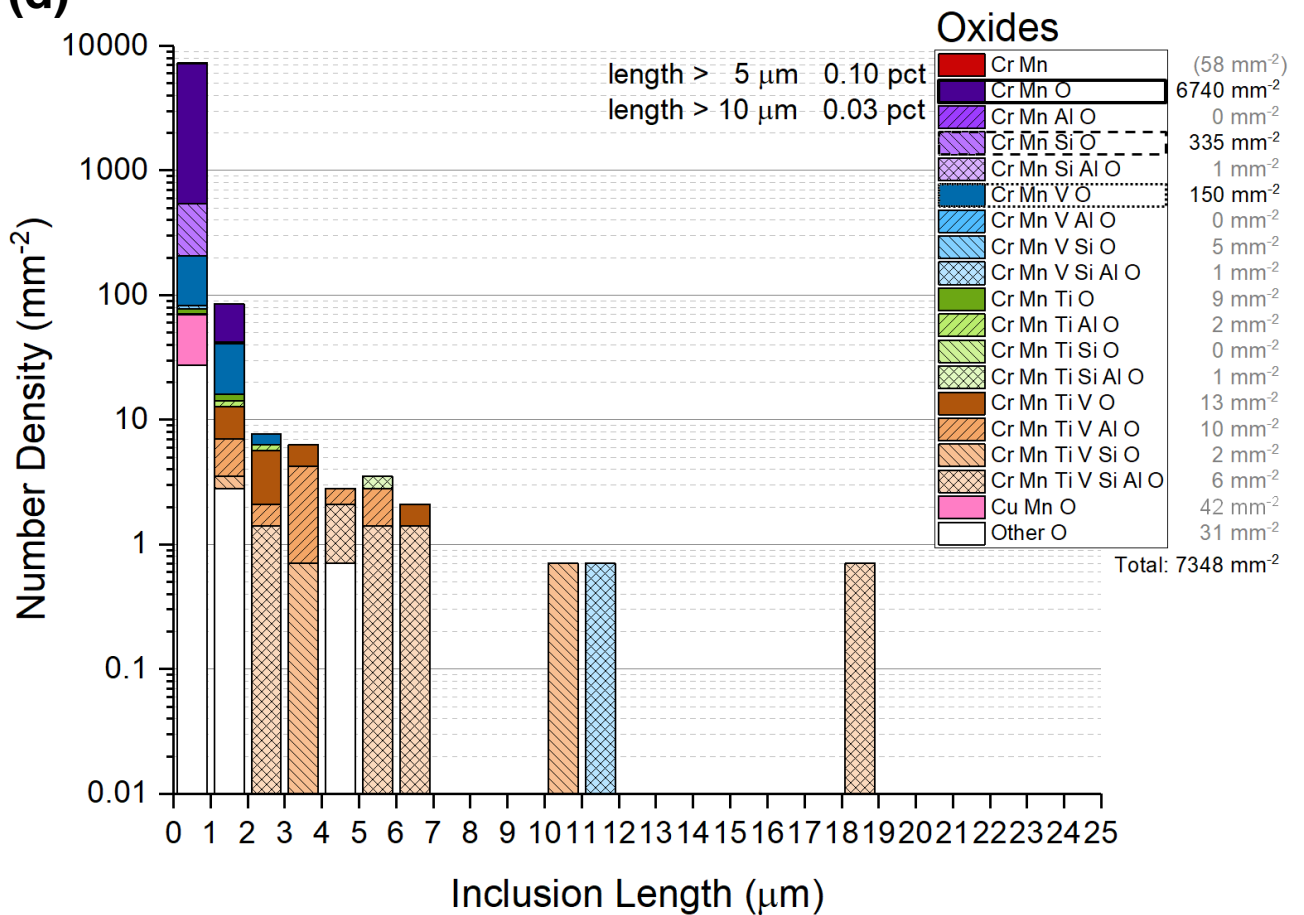
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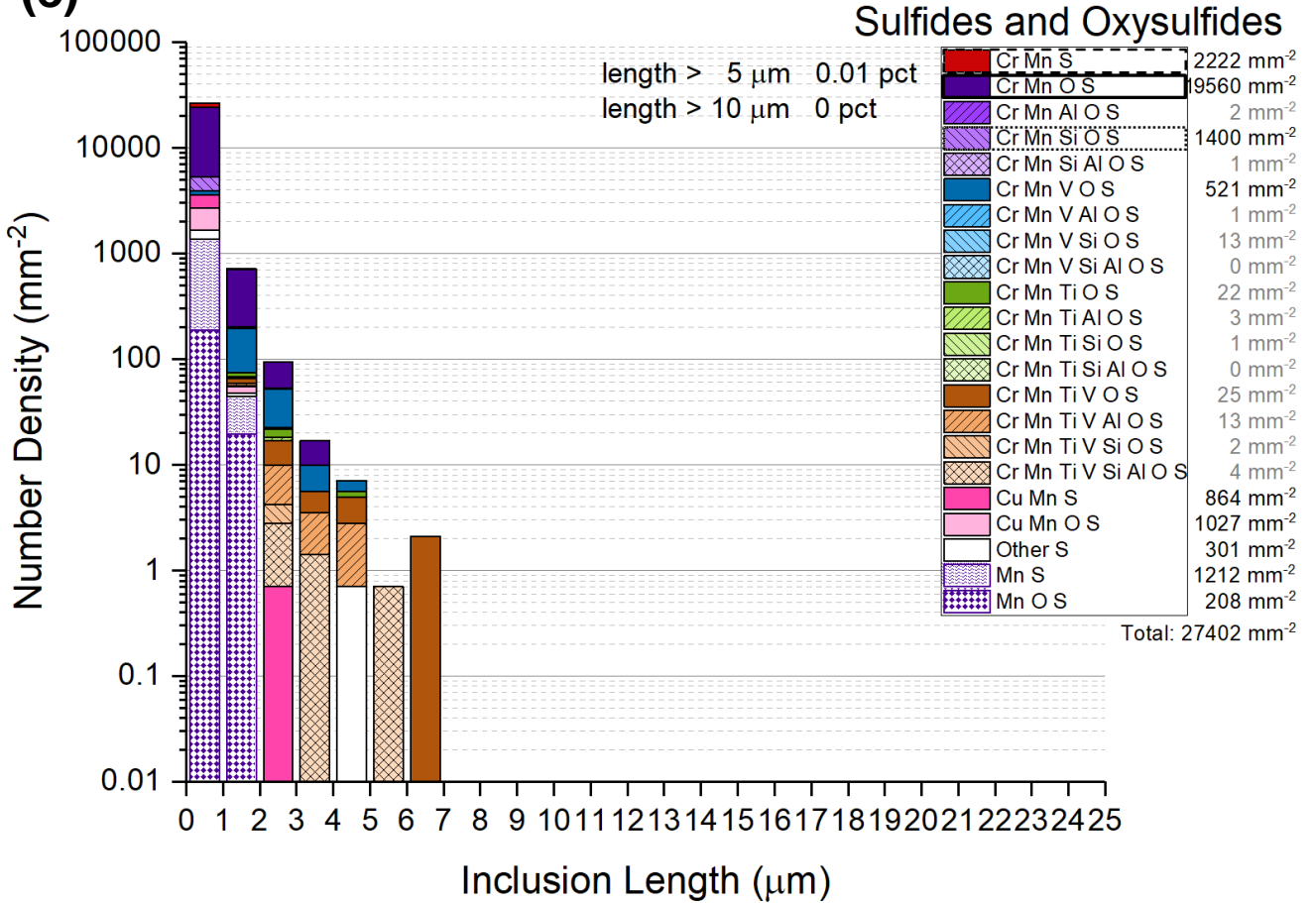
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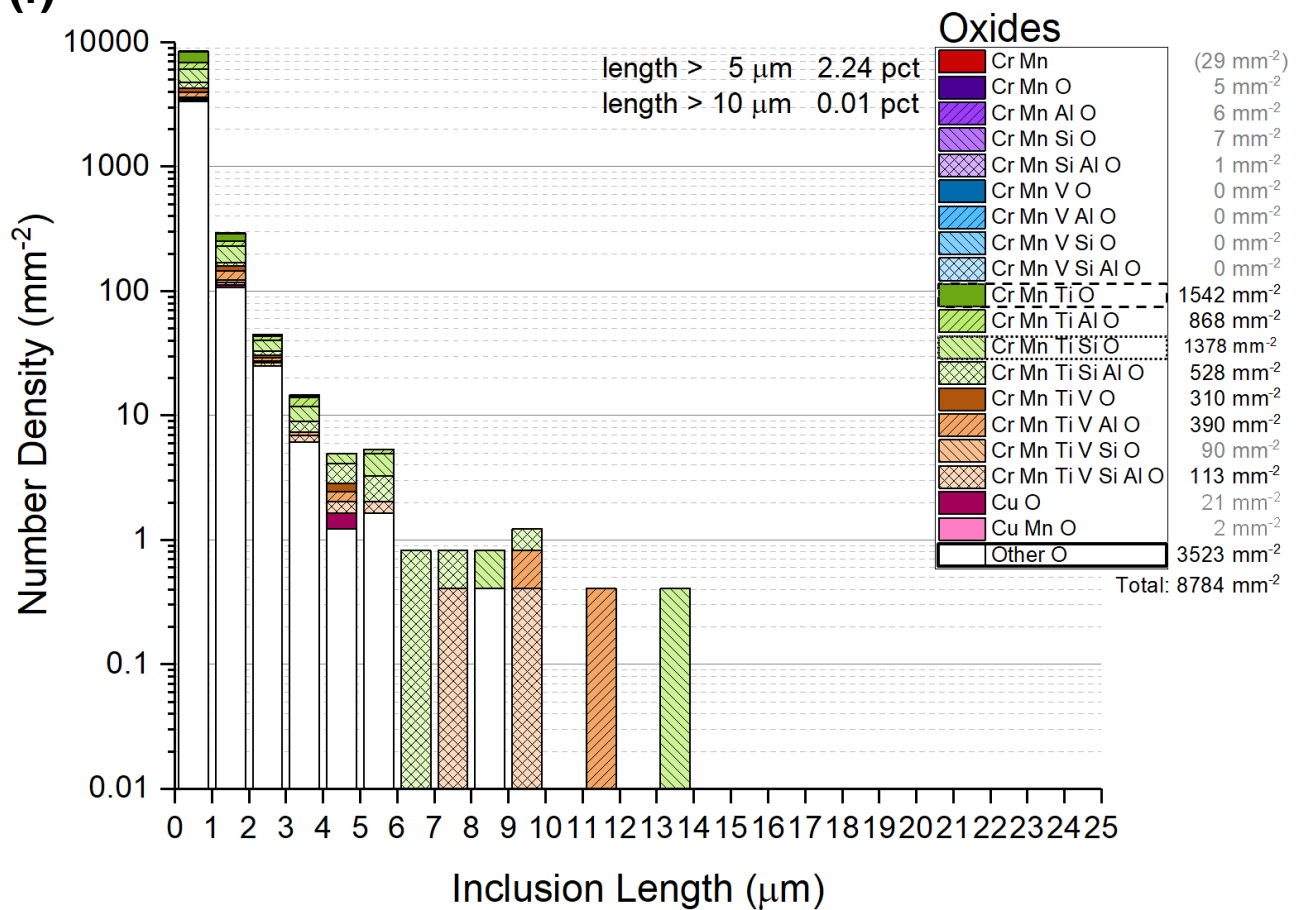
(d)



(e)



(f)



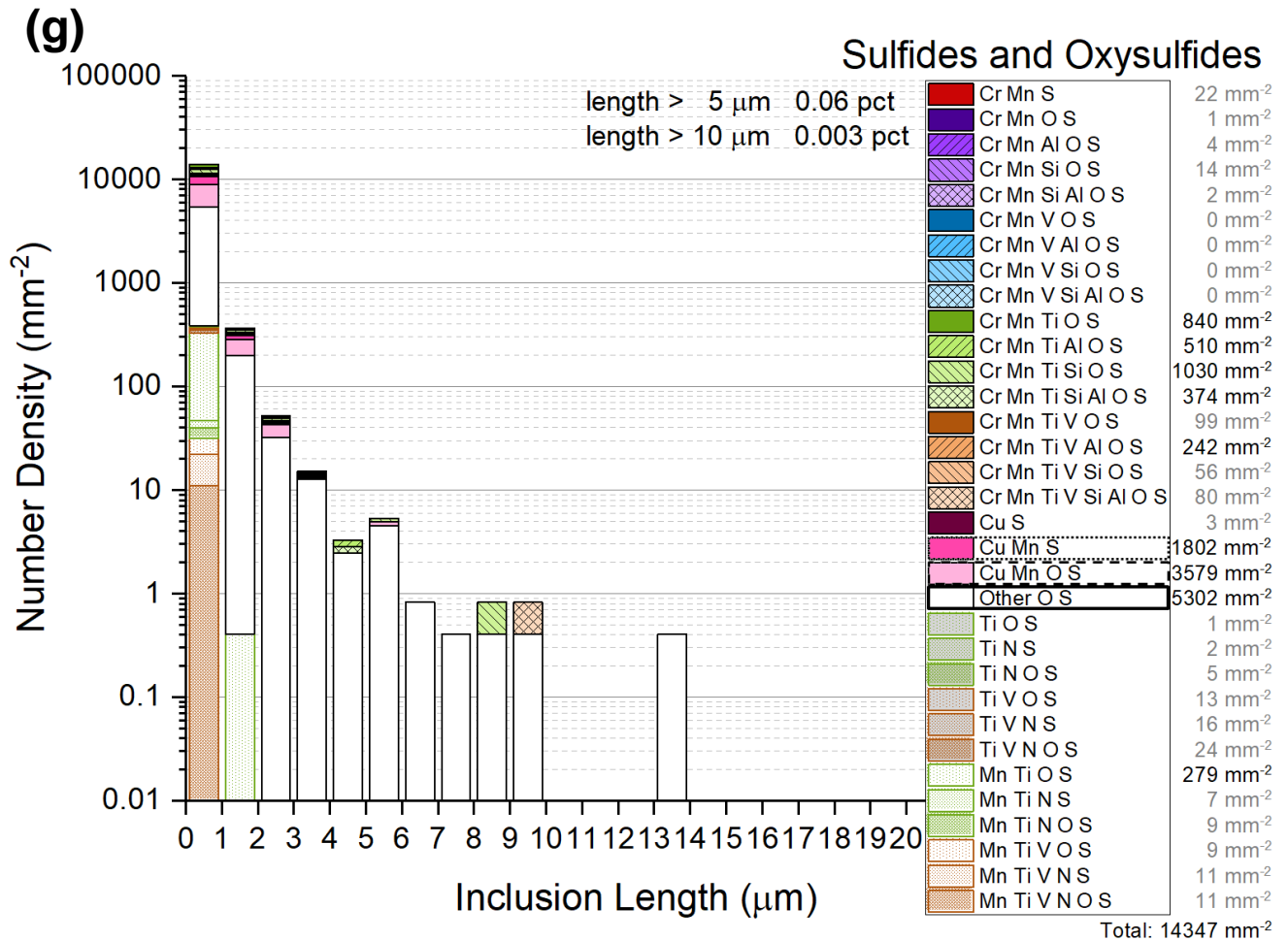
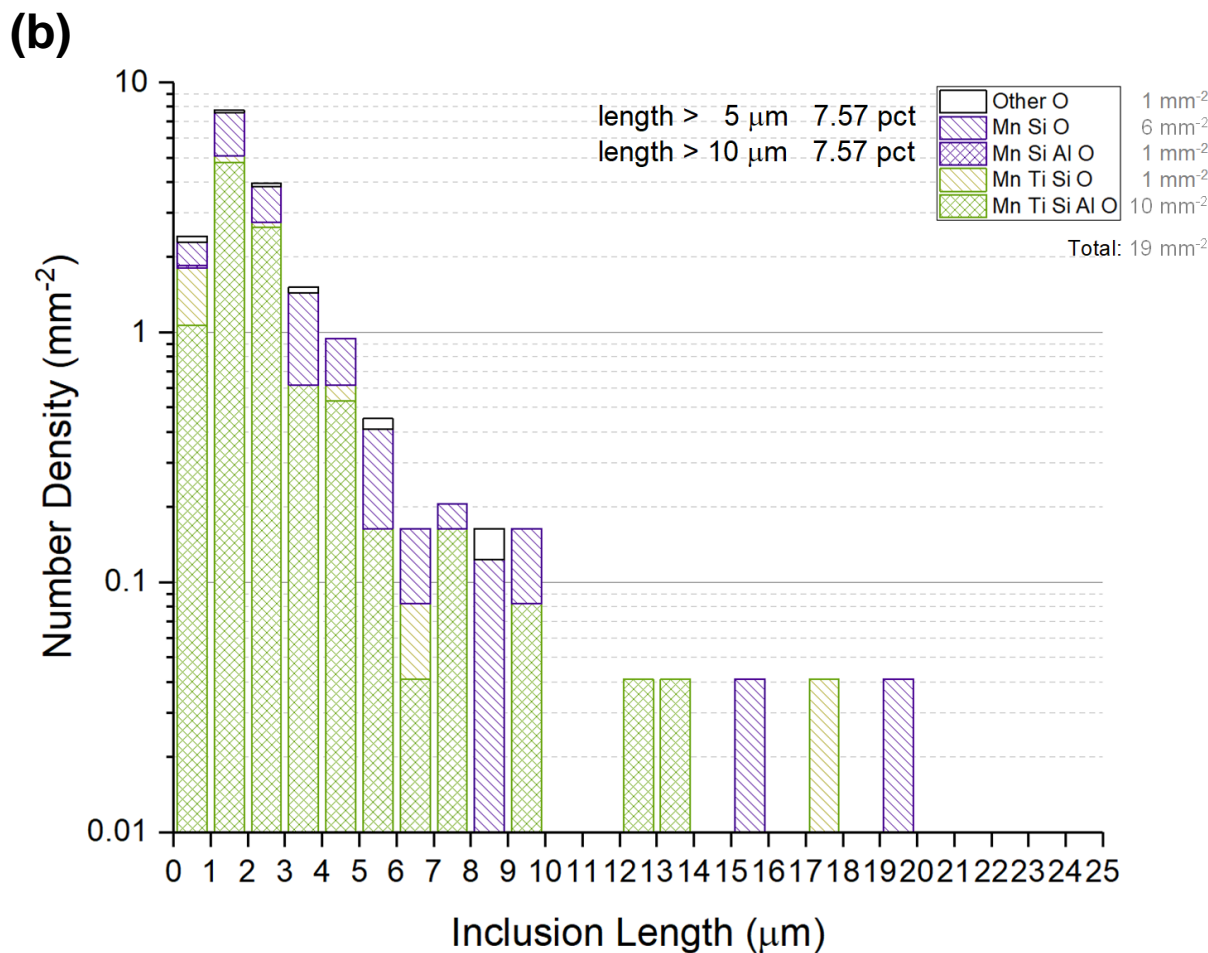
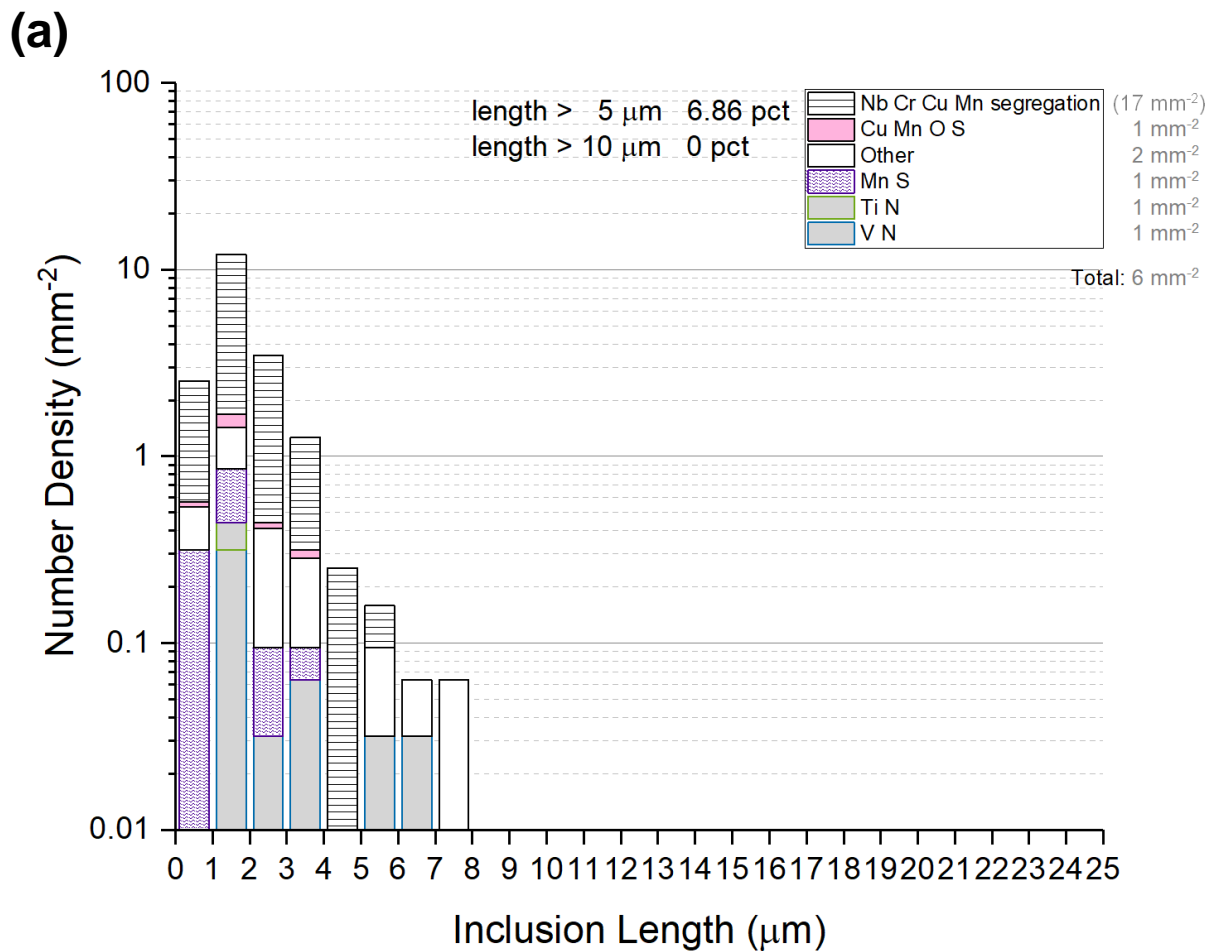
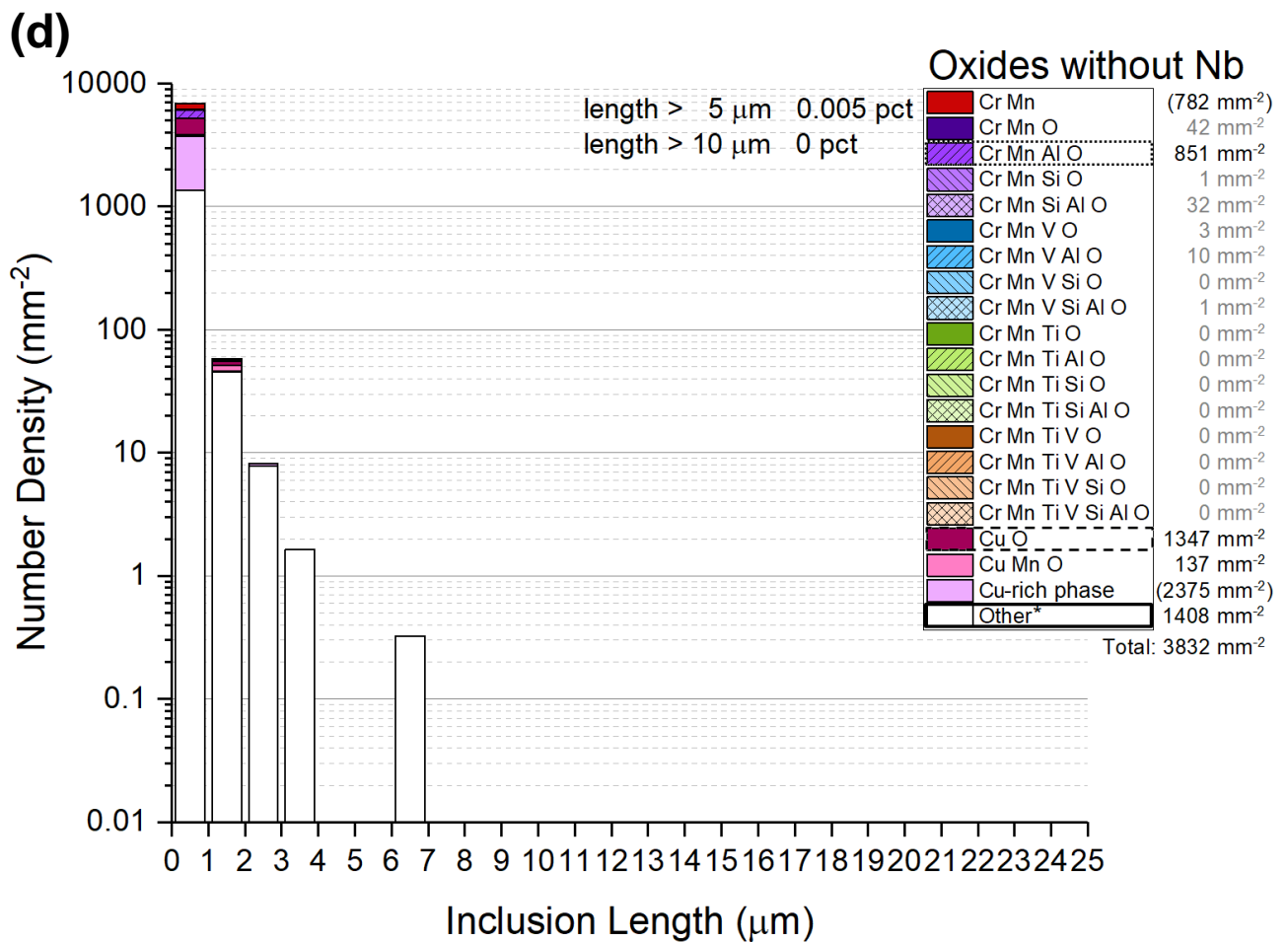
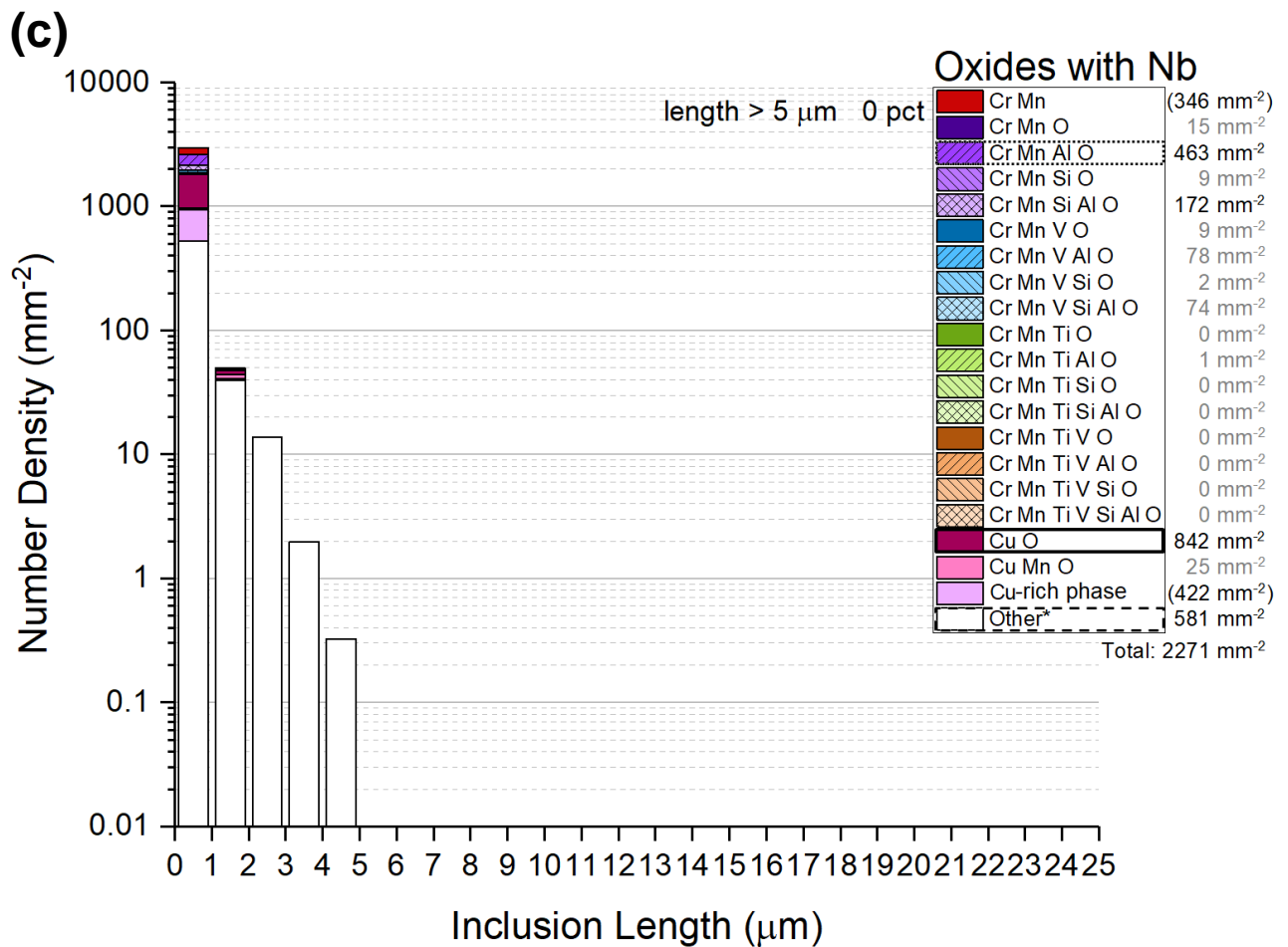
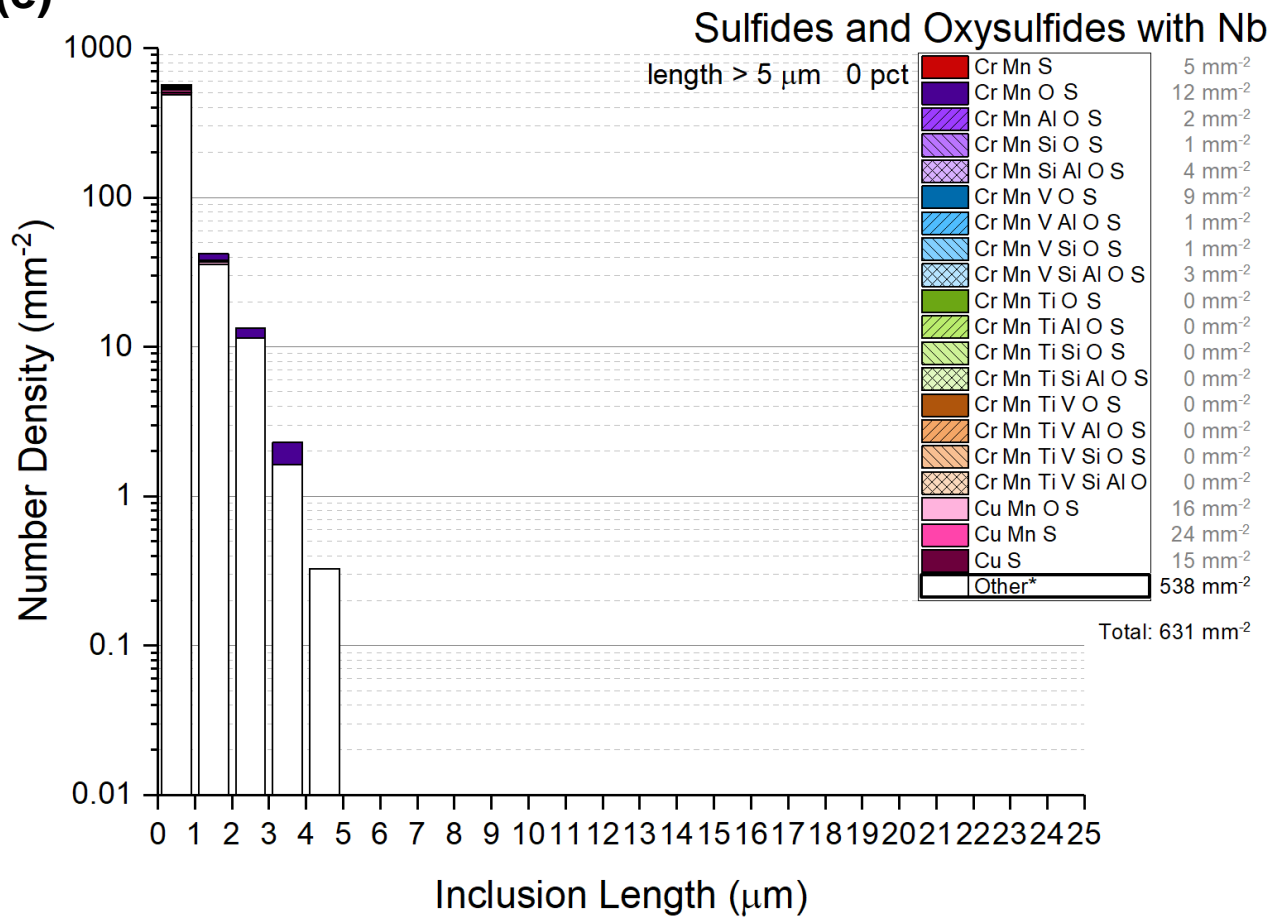


Fig. S-1 – Histograms of number density of non-metallic inclusions for the 316L stainless steel: (a) feedstock; (b) AMGA powder < 150 μm ; (c) VIGA powder 106 to 150 μm ; (d) and (e) AMGA HIP'd < 150 μm ; and (f) and (g) VIGA HIP'd 106 to 150 μm samples. The top three categories are marked in descending order of their value by continuous, dashed and dotted boxes, respectively. Number density values less than 100 mm^{-2} have been marked in grey color. Note that Si- and Al-containing inclusions are marked with fill patterns: hatched towards the left denotes Si, hatched towards the right denotes Al and hatched towards both the left and the right denotes Si and Al.





(e)



(f)

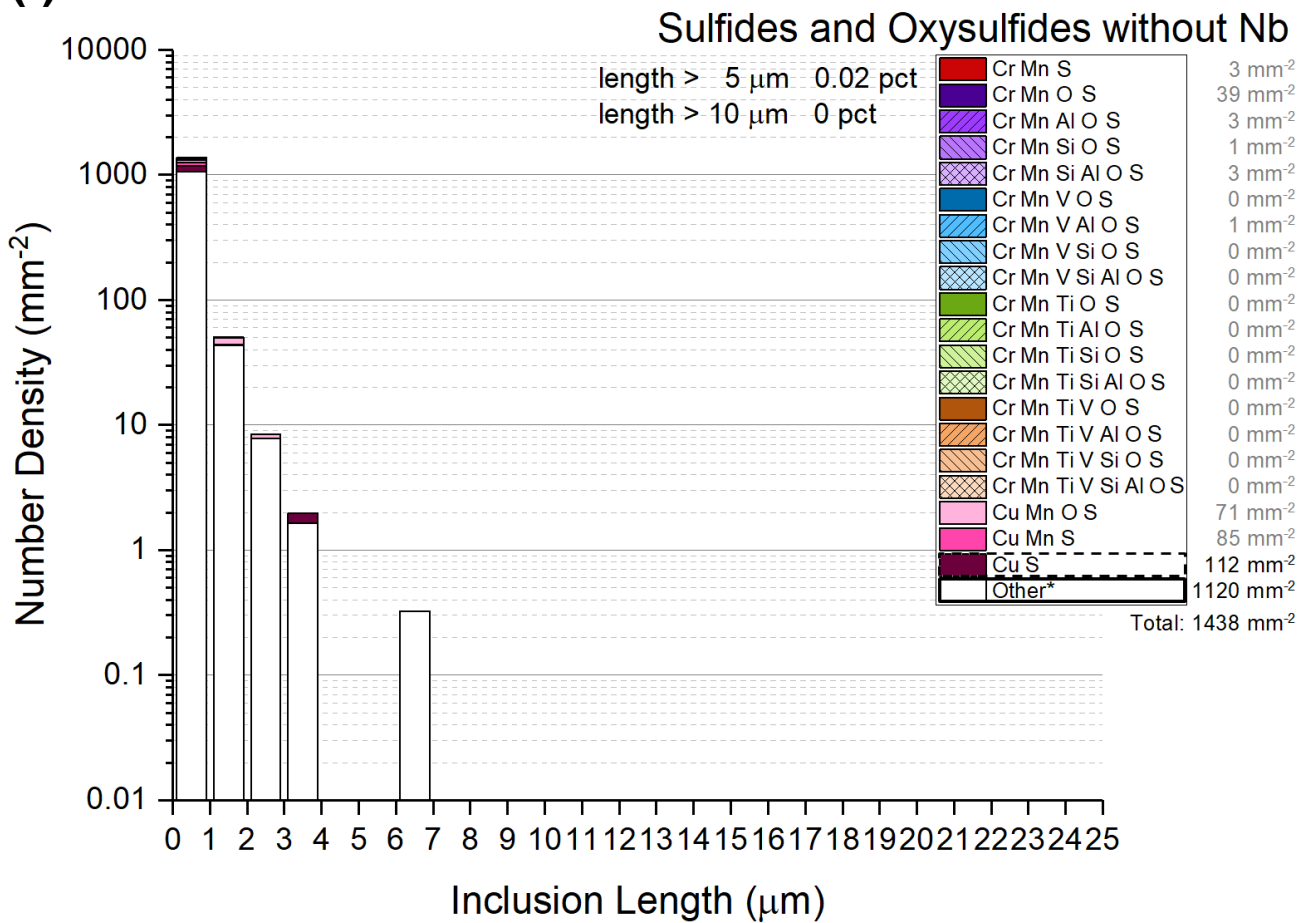


Fig. S-2 – Histograms of number density of non-metallic inclusions for the 17-4PH stainless steel: (a) feedstock; (b) VIGA powder < 150 μm ; and (c), (d), (e) and (f) HIP'd 25 to 45 μm samples. See Fig. S-1 caption for notations. *Other inclusions having Cr levels less than the matrix value of 15.78 wt pct, they are mostly Al-rich.

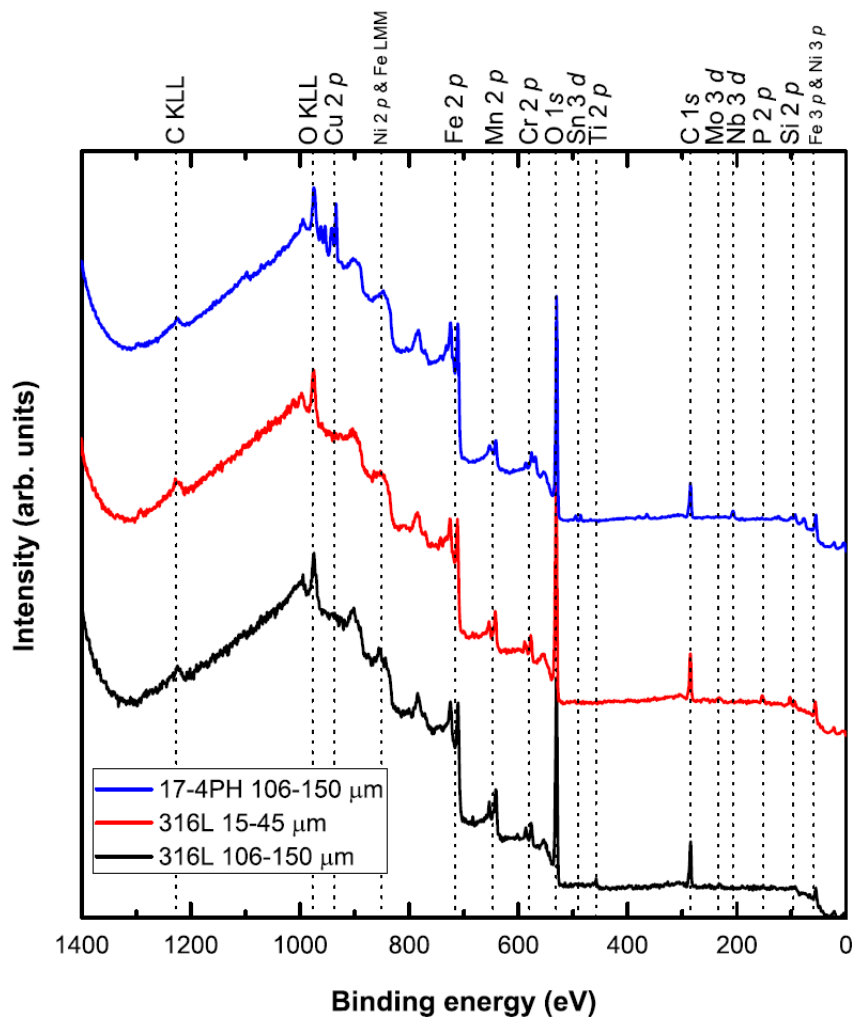


Fig. S-3 – Survey spectra acquired in the as-received condition from the three powders. The dotted lines indicate the approximate peak positions of the elements observed.

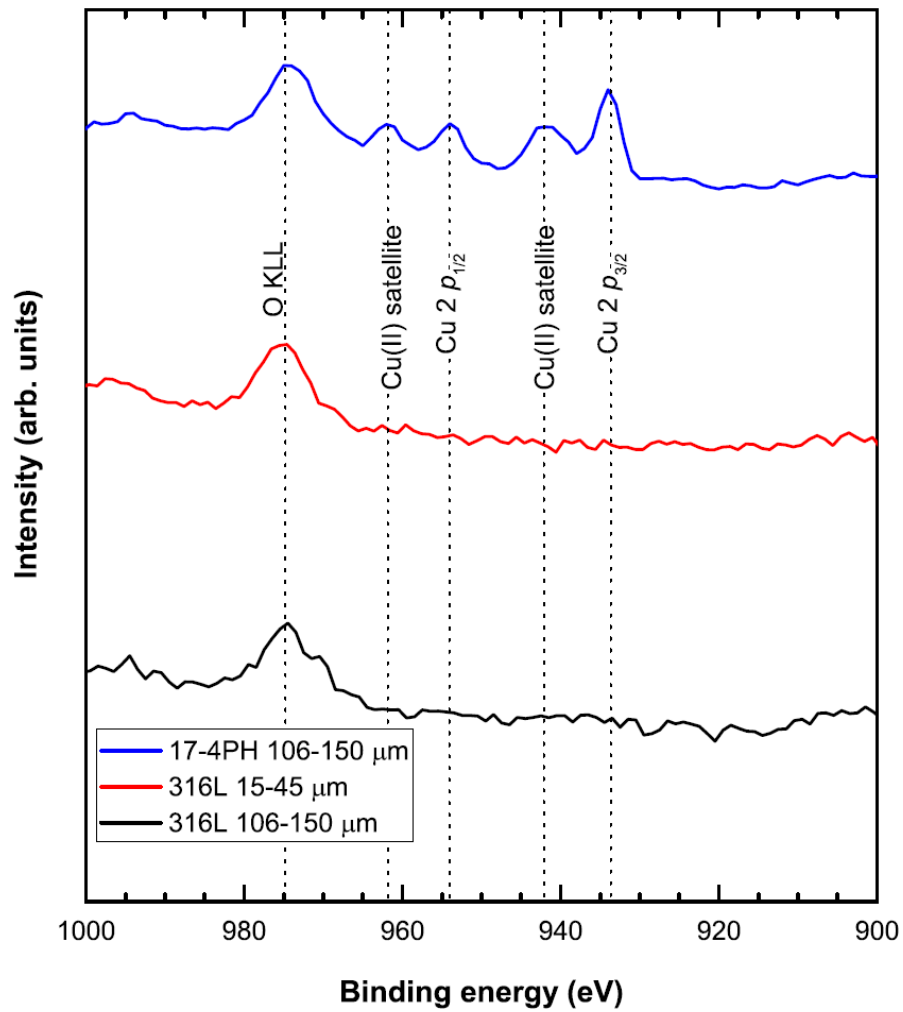


Fig. S-4 – The Cu 2p and O KLL region of the spectra acquired in the as-received condition from the three powders, clearly illustrating that Cu was only detected from the surface of 17-4PH 106 to 150 μm.

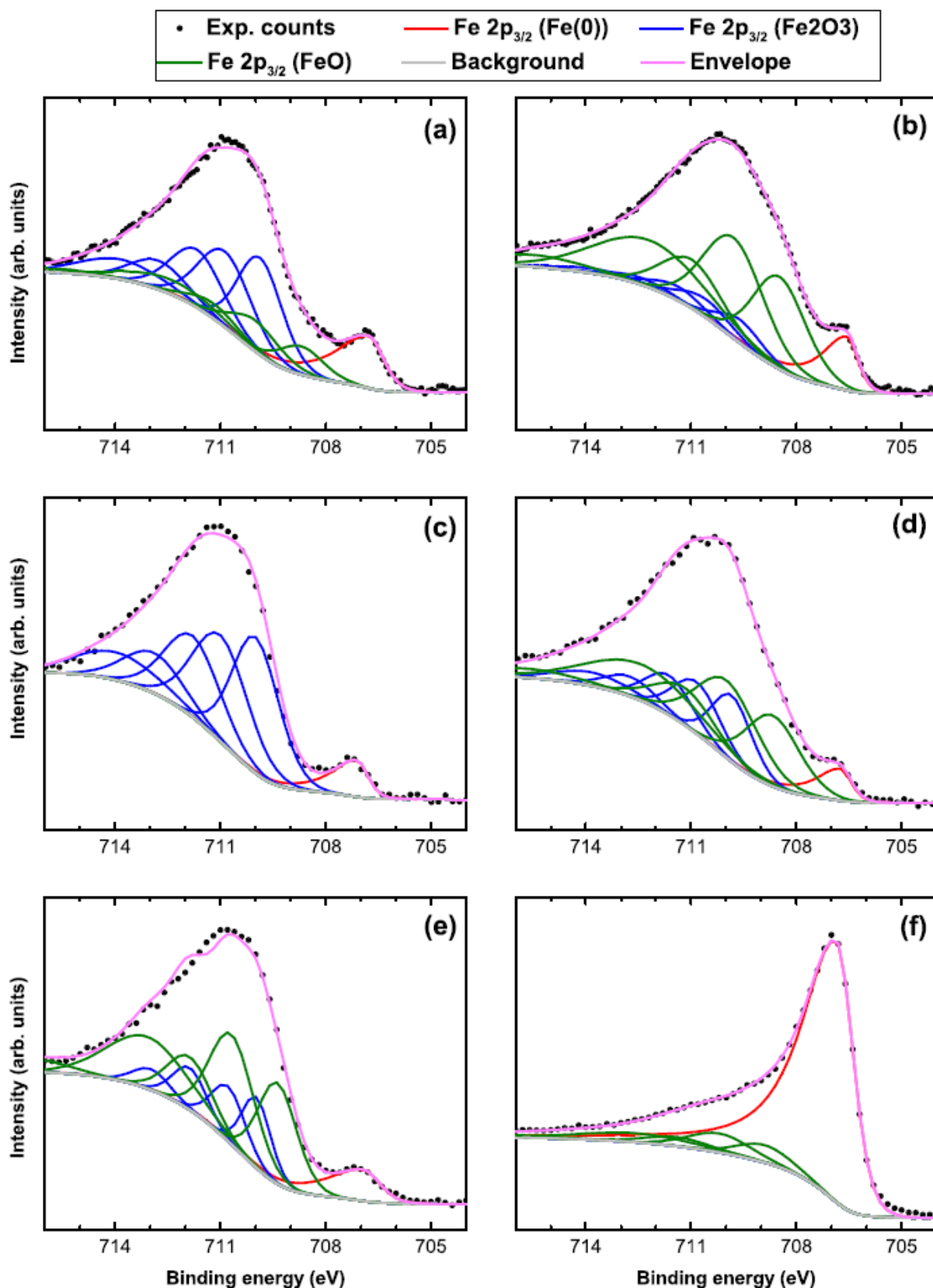


Fig. S-5 – Fe 2p_{3/2} spectra acquired from the three powders: 316L 105 to 160 μm (a) as-received and (b) after sputtering; 316L 15 to 45 μm (c) as-received and (d) after sputtering; 17-4PH 106 to 150 μm (e) as-received and (f) after sputtering. Contributions from metallic Fe(0), FeO and Fe₂O₃ are seen throughout, as detailed in Table XI.

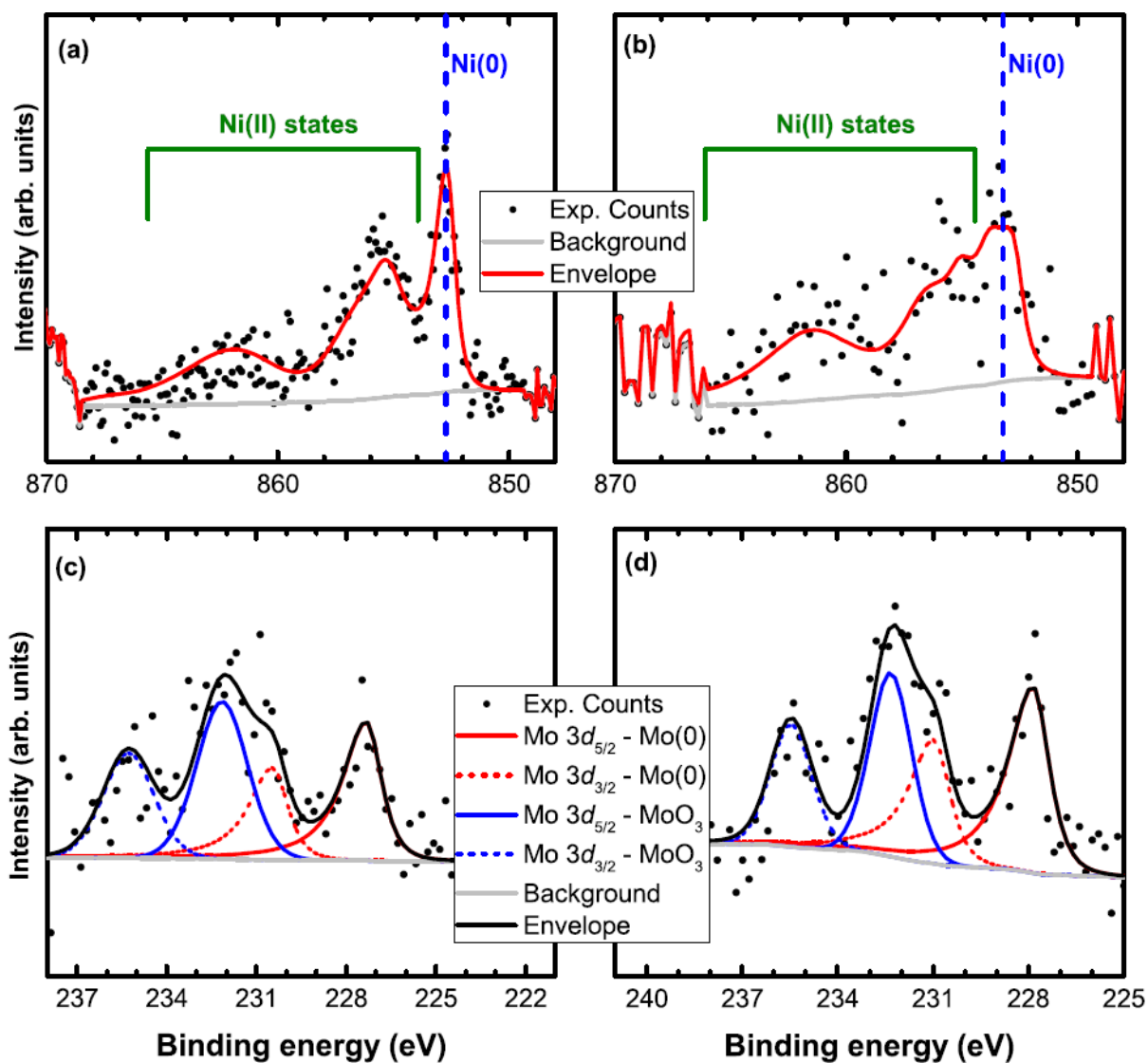


Fig. S-6 – High resolution core level spectra examining the Ni 2p region (a) and (b); and the Mo 3d region (c) and (d) for the presence of surface oxides. Both as-received 316L 106 to 150 μm (a) and (c); and the 316L 15 to 45 μm (b) and (d) presented evidence of Ni and Mo oxides near the surface. 17-4PH and did not have Ni or Mo in sufficient quantity near the surface to examine this effect.

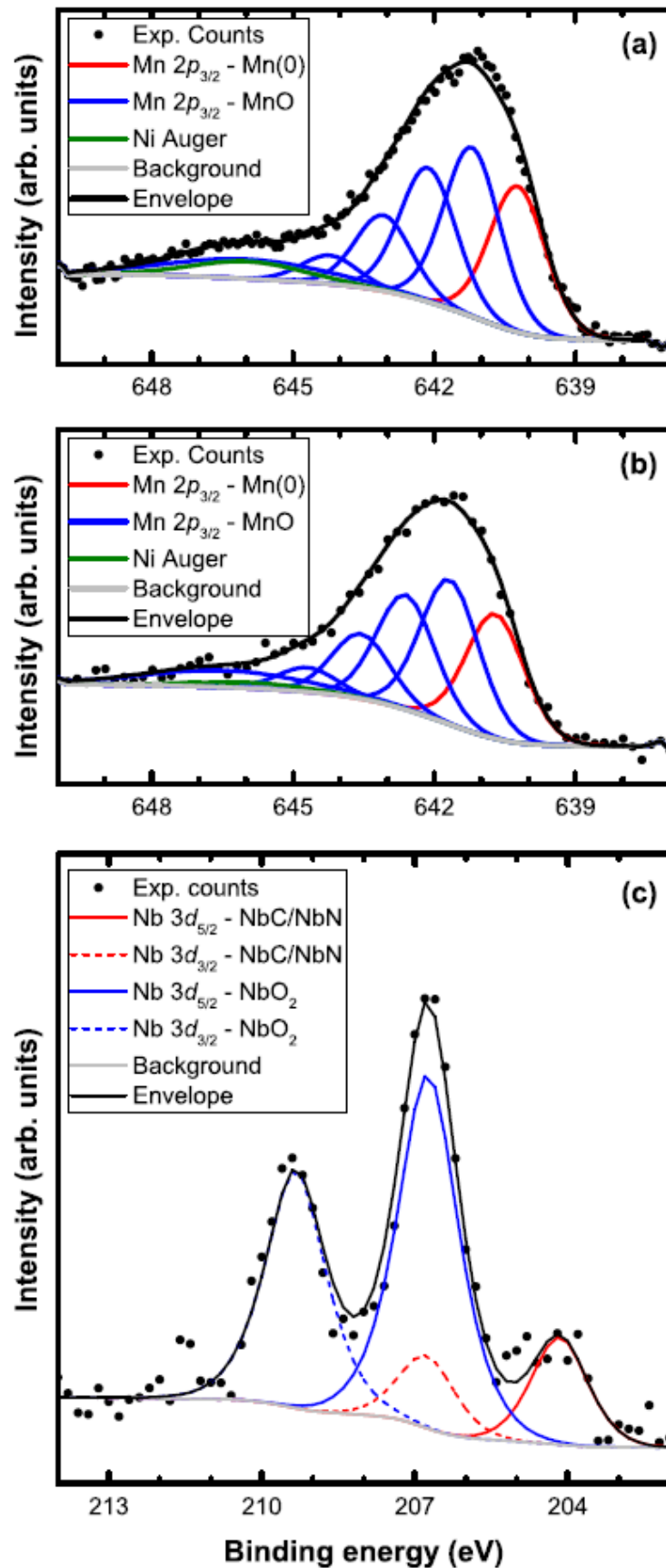


Fig. S-7 – Mn $2p_{3/2}$ spectra from (a) the 316L 106 to 150 μm powder and (b) the 316L 15 to 45 μm powder, showing the existence of MnO near the surface. Panel (c) shows the Nb $3d$ spectrum from the 17-4PH powder, with contributions from oxides and carbides and/or nitrides.

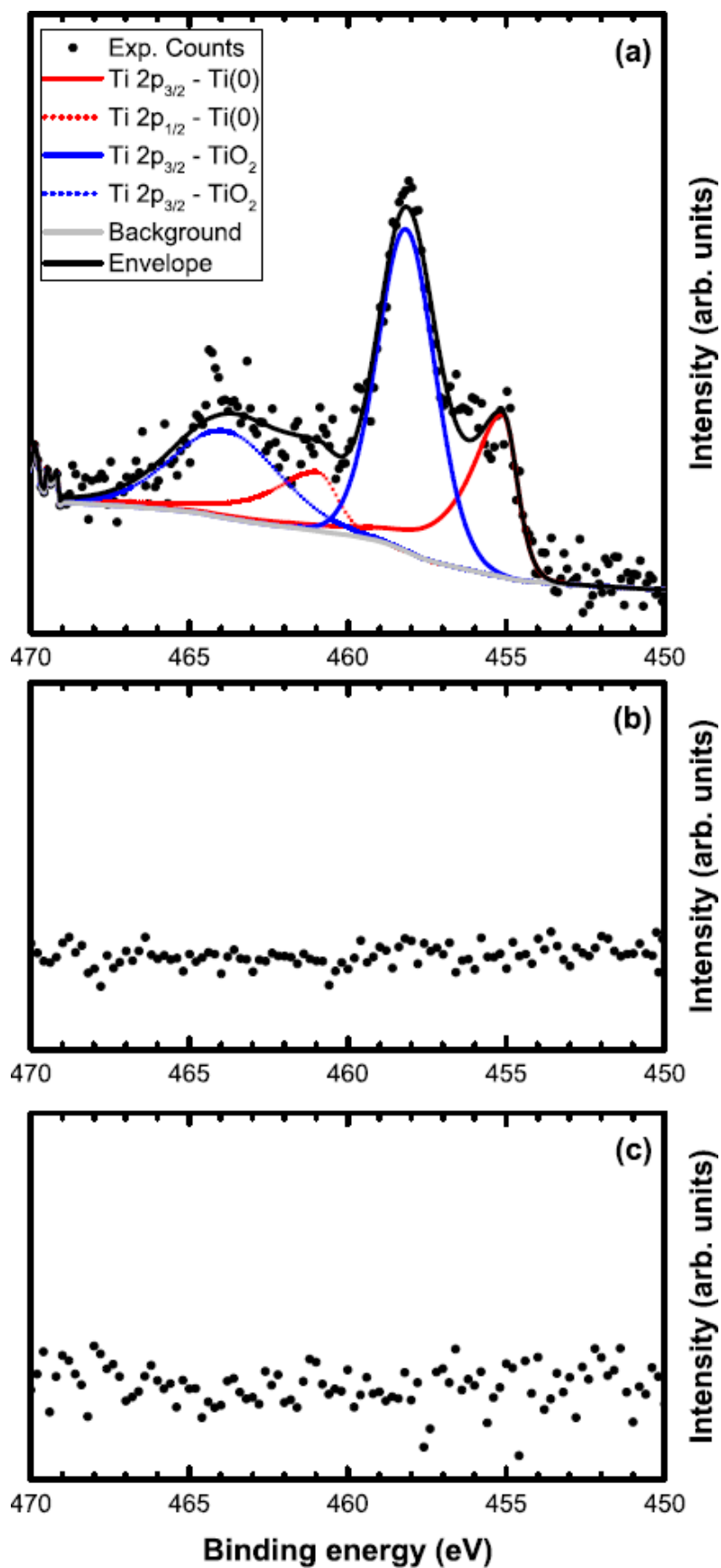


Fig. S-8 – Ti 2p spectra from (a) the 316L 106 to 150 μm powder, (b) the 316L 15 to 45 μm powder, and (c) the 17-4PH powder. Ti was only observed on the 316L 106 to 150 μm powder, with contributions from both metallic and oxidized Ti.

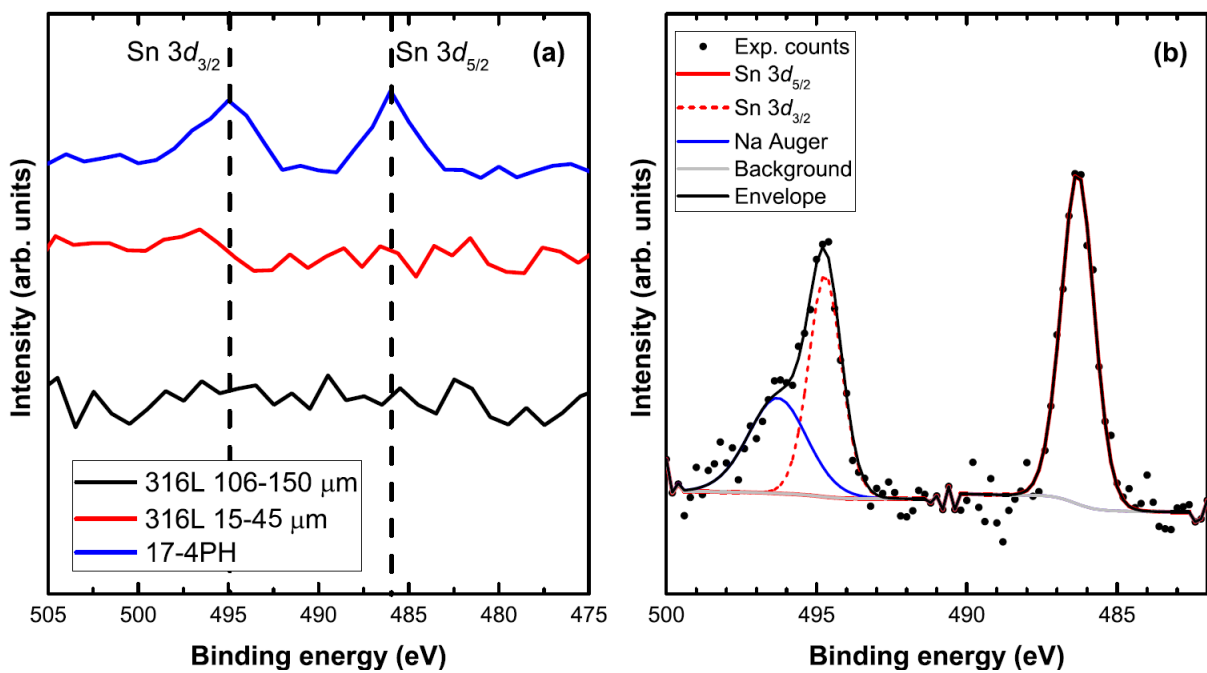


Fig. S-9 – (a) The Sn 3d region extracted from the survey spectra acquired from each of the three steel powders, showing that Sn was only detected in 17-4PH and (b) The high-resolution Sn 3d region from 17-4PH, showing the existence of Sn oxides on the surface.

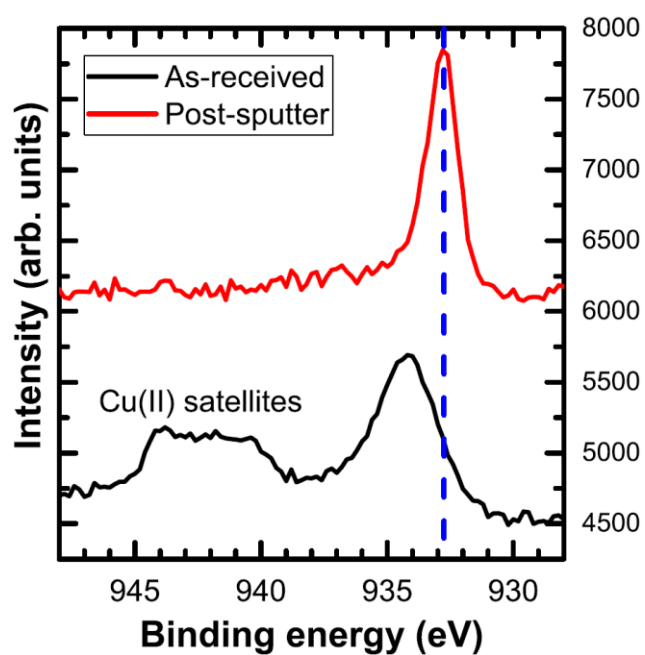


Fig. S-10 – The Cu $2p_{3/2}$ region acquired from the 17-4PH powder in the as-received condition and after Ar^+ sputtering. The initial spectrum is dominated by Cu(II) components, but these are removed to reveal a Cu_2O layer and underlying metallic Cu. The blue dashed line indicates the binding energy of the main Cu $2p_{3/2}$ component after sputtering, highlighting the downward energy shift of the envelope due to the removal of Cu(II) species.

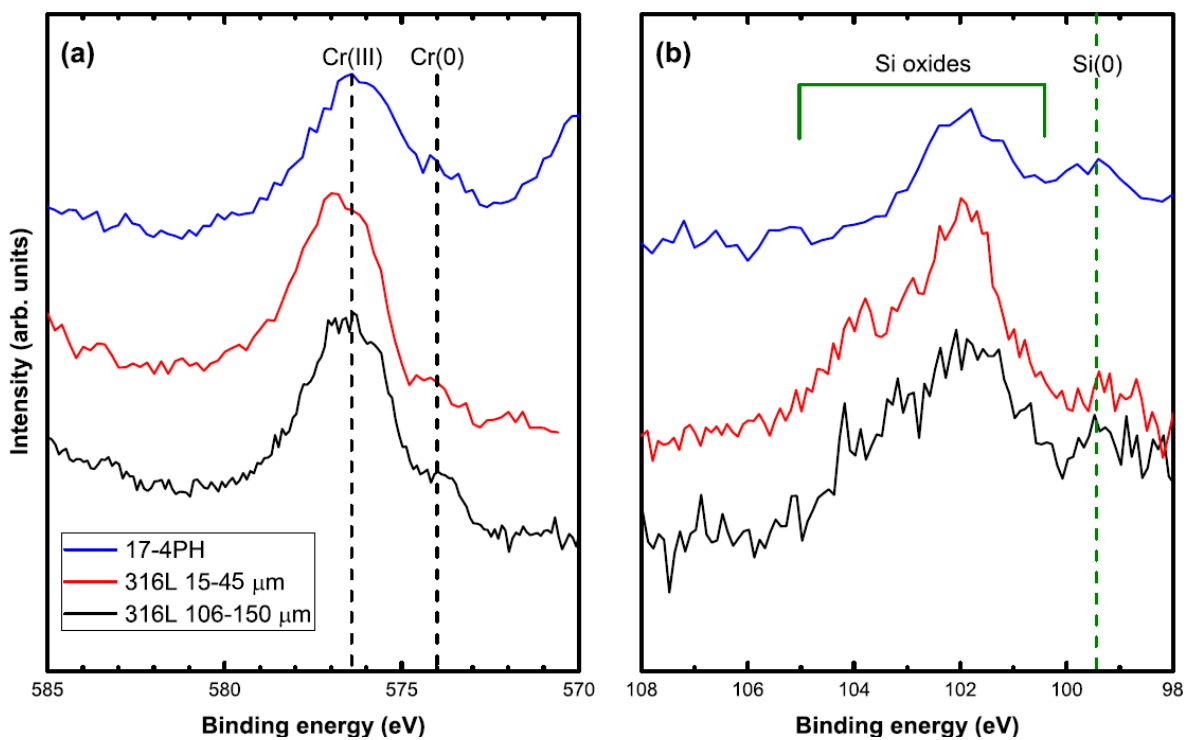


Fig. S-11 – (a) The Cr $2p_{3/2}$ spectra acquired from the three powders in the as-received condition, showing the presence of both Cr(0) and Cr(III) species and (b) The Si $2p$ spectra acquired from each powder in the as-received condition, dominated by Si oxides but also showing a small contribution from Si(0).

Grain Size in the HIP'd 316L Steel

Grain size distributions (determined as equivalent grain diameter as the grains are approximately equiaxed) are plotted in supplementary Fig. S-12 and the corresponding average and standard deviation are also given in supplementary Fig. S-12. It can be seen that the as-atomized conditions exhibit much finer grain size microstructures compared to their HIP'd products, confirming grain growth during the HIPping and heat treatment process. The average and standard deviation grain size values in the AMGA HIP'd 316L < 150 μm were somewhat coarser ($23.1 \pm 21.6 \mu\text{m}$) than those of the VIGA HIP'd 316L 106 to 150 μm ($17.3 \pm 15.4 \mu\text{m}$). The AMGA HIP'd 316L < 150 μm sample would be expected to give a finer grain structure given the number density of non-metallic inclusions of the HIP's samples, but this is the opposite trend to that shown here. It is, however, of interest to note that Ti was partly retained in the VIGA HIP'd 316L 106 to 150 μm sample (0.05 wt pct), compared to the AMGA HIP'd 316L < 150 μm sample (< 0.005 wt pct). Grain refinement would then result from the presence of strong grain boundary pinning centers, e.g. Ti-rich carbonitrides, as some grain boundary pinning from precipitates has been observed in both HIP'd samples (Figures 10 and 11). Furthermore, a bimodal grain size is seen for the HIP'd 316L < 150 μm (air melted) material with 0.085 area fraction of 130 - 140 μm sized grains.

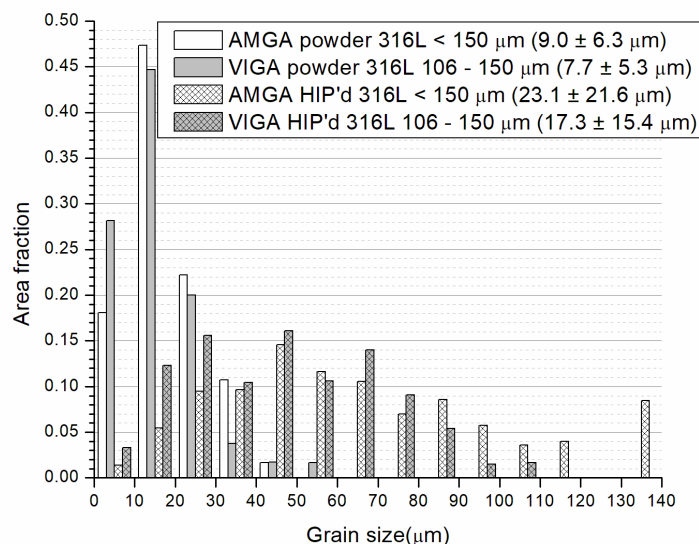


Fig. S-12 – Grain size distributions as measured by equivalent diameter in powders and after the HIPping and heat treatment process.

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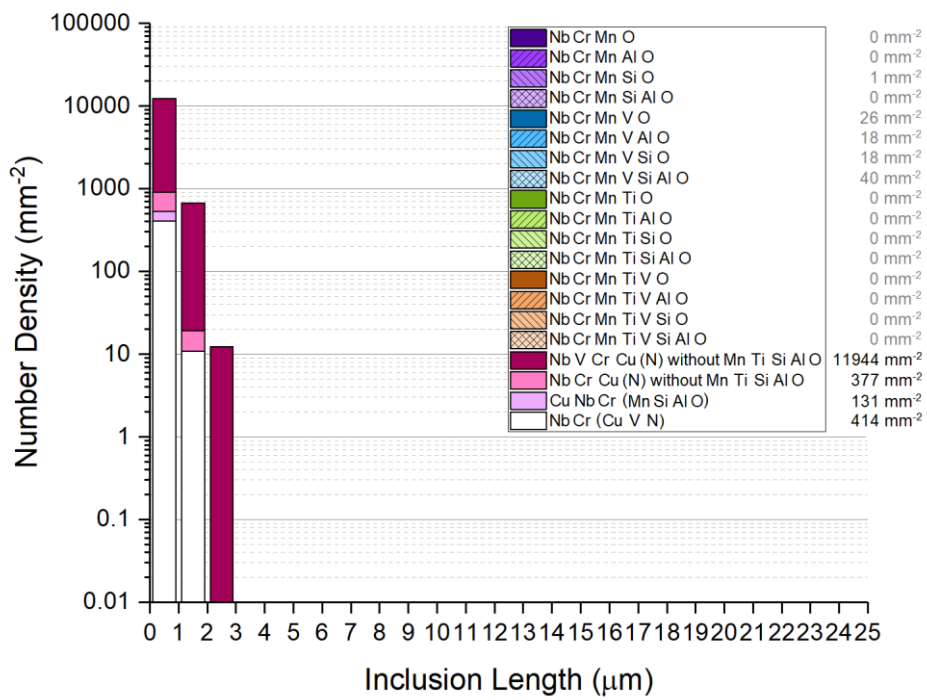


Fig. S-13 – Histograms of the number density of the white contrast phase in BSEI for the HIP'd 17 - 4PH 25 to 45 μm sample. Elements in parenthesis indicate that can sometimes be present or not.